

LOW TEMPERATURE RHEOLOGICAL AND FRACTURE PROPERTIES OF POLYMER-MODIFIED BITUMENS

**L.Lapalu¹, J.P. Planche¹, D. Martin¹, D. Anderson², J.F.
Gérard³**

¹ ELF Research Center (Solaize)

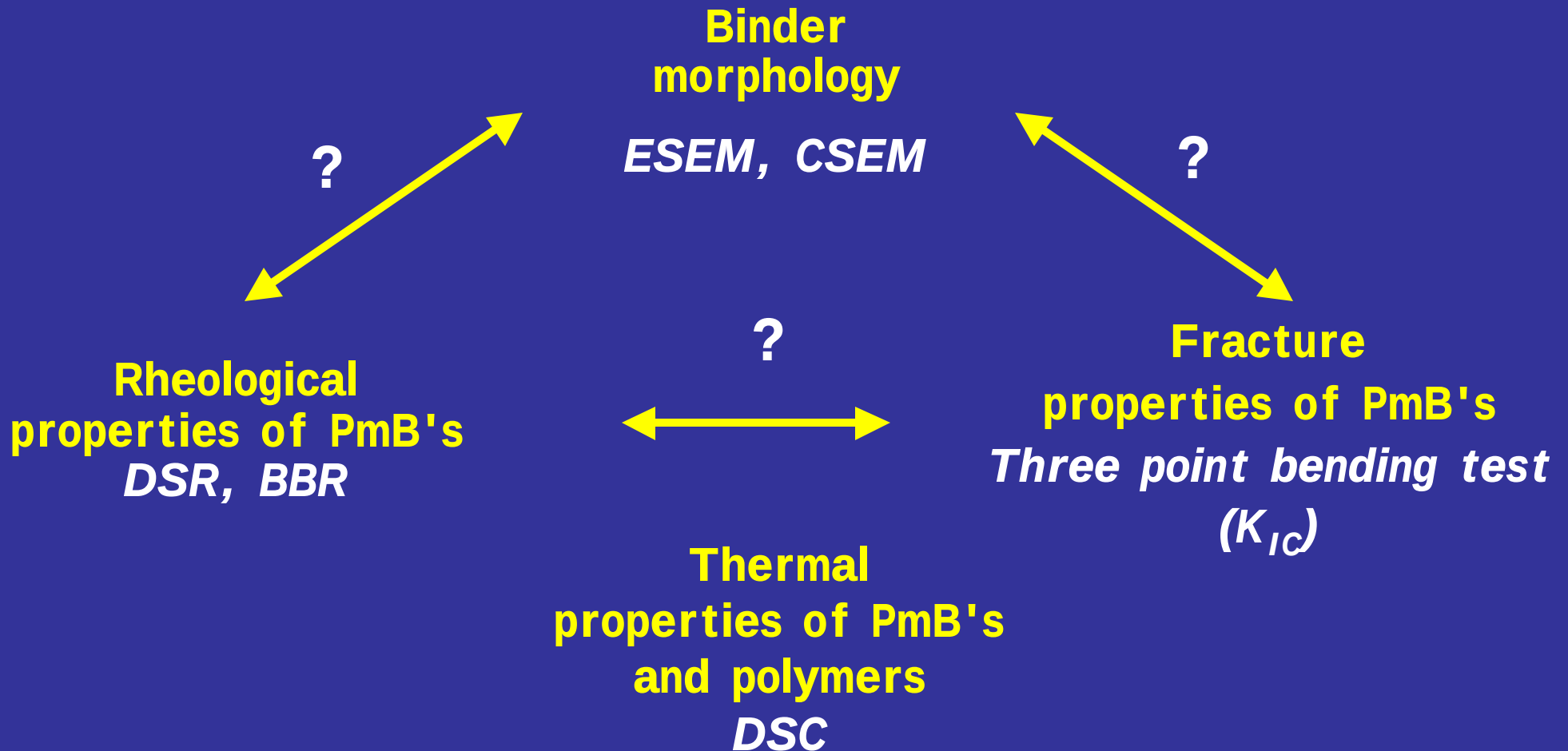
² The Pennsylvania State University

³ National Institute of Applied Science (Lyon)

Eurasphalt & Eurobitume Congress

Barcelona, September 2000

THE AIM OF THE STUDY



Analyze of the effects of the polymer chemical nature, polymer

MATERIALS

Neat bitumen

Grade :	70/100
Penetration at 25°C (1/10 mm) :	85
Ring & Ball softening point (°C) :	46
Tg (°C) :	-27
% Crystallized fractions (DSC) :	4.5

MATERIALS

Semi-crystalline Copolymers : Plastomers

Copolymer	Fraction of the comonomer (% wt.)	T _g (°C)	Crystallinity %
EVA - 18	18.6	-22	25.7
EVA - 28	28.4	-20	15.3
EMA - 28	28.6	-26	9.4
EBA - 35	33.9	-46	10.6

PmB's produced by mixing the base bitumen with 4 to 6 weight % semi-crystalline copolymers

All physical blends

MATERIALS

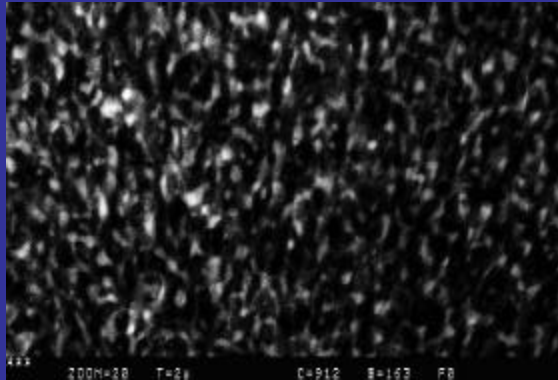
Amorphous copolymers : Elastomers

Copolymer	Styrene content (% wt.)	T _g (°C) (PB / PS blocks)	M _w (dalton)
SBS ₁	30	- 90 / 51	240 000
SBS ₂	30	- 90 / 62	135 000
SBS ₃	18	-100 / 66	250 000
SB	18	-88 / 66	250 000

PmB's produced by mixing the base bitumen with 4 weight % amorphous copolymers

Four physical blend - One chemically in-situ crosslinked blend

OBSERVATION OF THE BLEND MORPHOLOGY USING CLSM

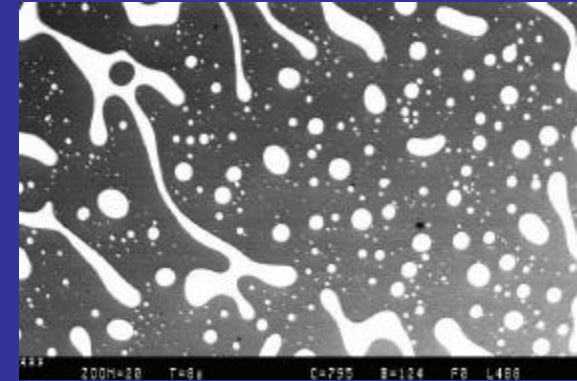


Blend based on 4% wt.
EBA - 35

*co-continuous
morphology*

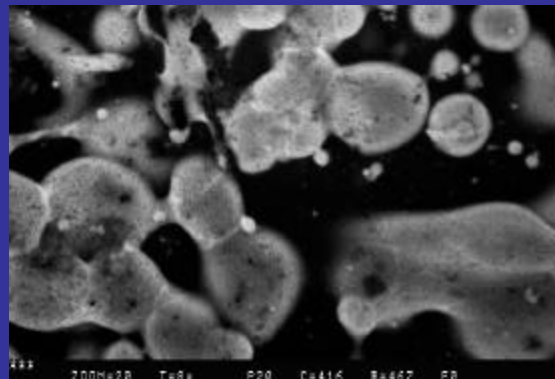
Blend based on 4% wt.
EMA - 28

*polymer-rich particles
dispersed in the
bitumen-rich matrix*



Blend based on 4% wt.
SBS*₃

*polymer-rich particles
dispersed in the
bitumen-rich matrix*



50 μ m
↔

MEASUREMENT OF RHEOLOGICAL PROPERTIES

DSR :

Rheometrics RDA II

Measurement at 1Hz from -120°C to 100°C

G' , $G'' = f(T)$

BBR :

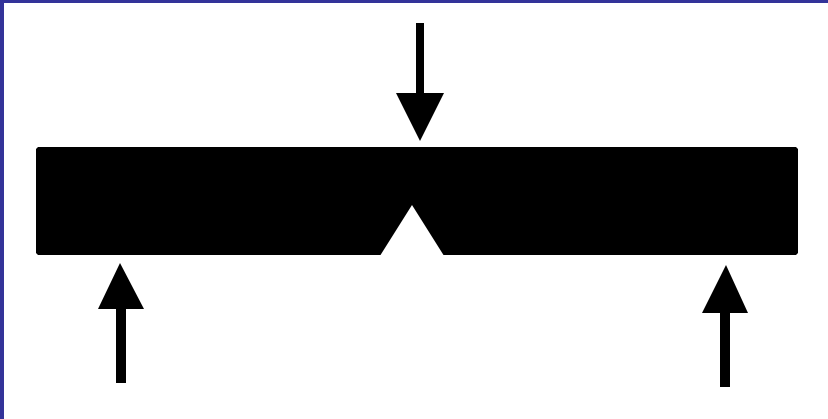
BBR Cannon Instrument Company

Tests performed at -18°C, -21°C and -24°C

Measurement of $T_{S=300\text{MPa}}$ and $T_{m=0.3}$

Tests run on unaged binders

THE MODE-I FRACTURE TEST



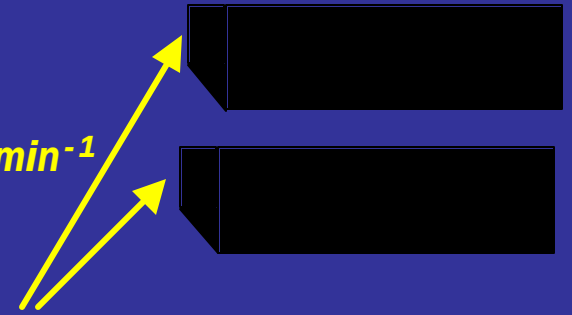
Environmental chamber at -20°C

Test



Speed = 0.6 mm.min⁻¹

T = -20°C

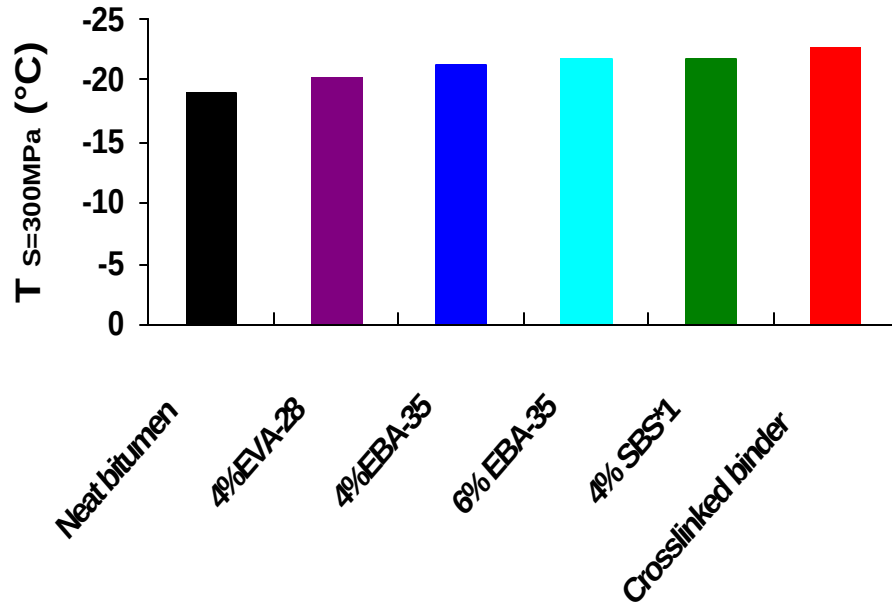
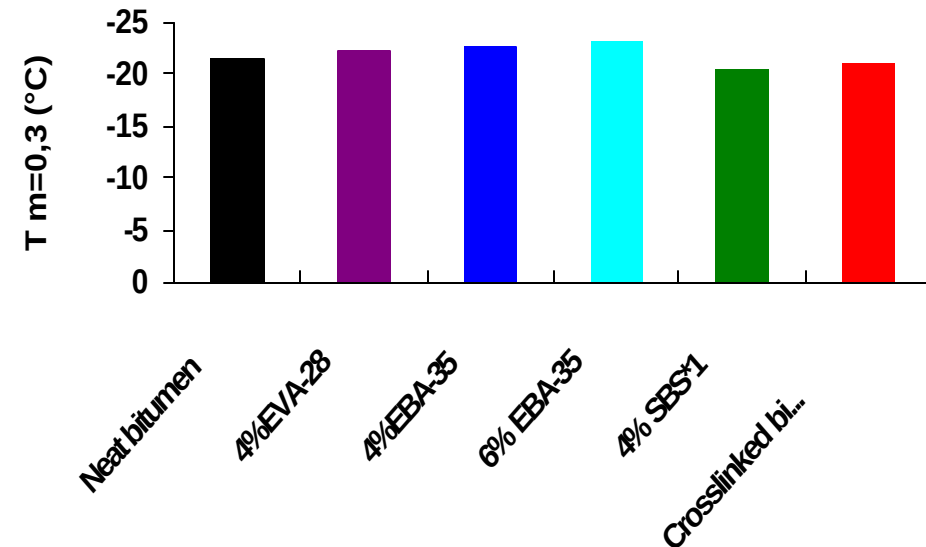


*Cutting of the rupture faces for
CSEM or ESEM observations*

K_{IC} = failure load $\times f$ (span, crack length, sample dimensions)

8 replicates necessary for each sample

RHEOLOGICAL PROPERTIES : BBR RESULTS



BBR results : Polymer addition imparts low improvement to binder creep properties

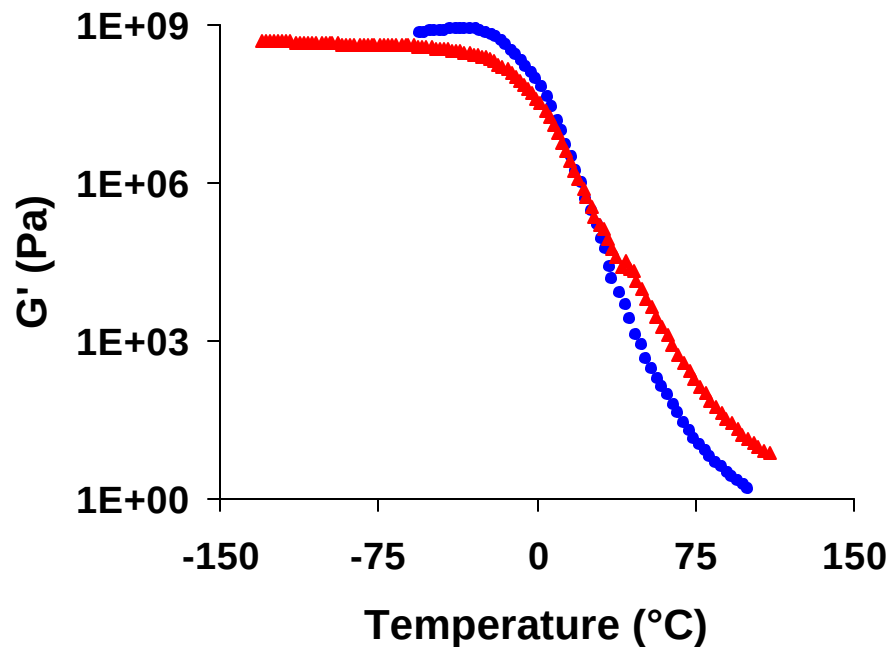
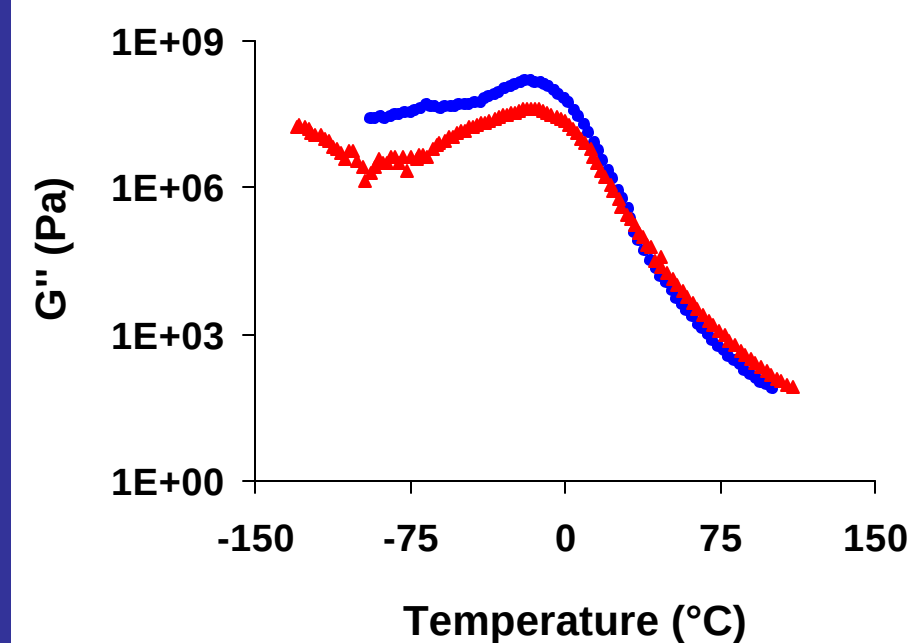
BBR does not discriminate binders with respect to their rheological properties at low temperature

RHEOLOGICAL PROPERTIES : DSR RESULTS

Improving the low temperature rheological properties by polymer modification should result in decreasing the elastic modulus G' and/or increasing the viscous modulus G''

*4% SBS*₃ Blend*

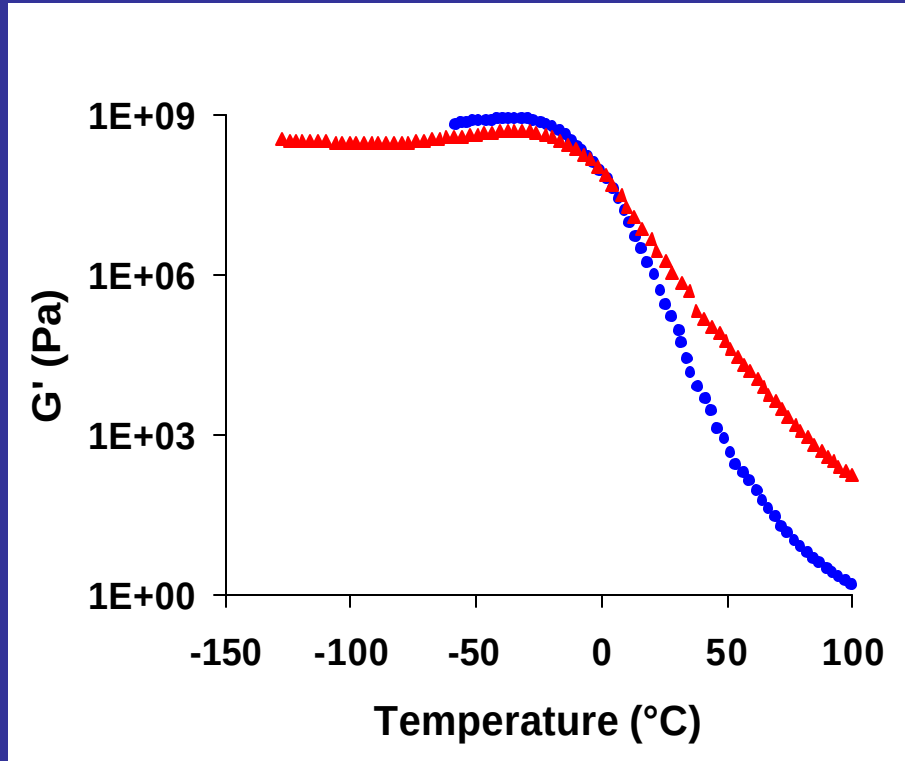
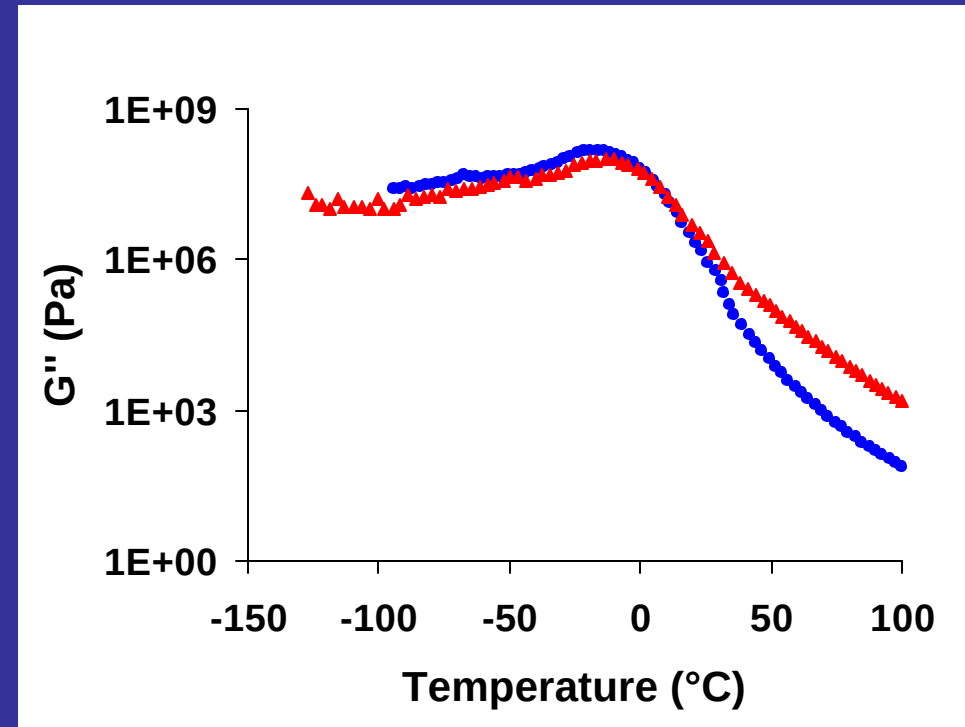
Neat bitumen



RHEOLOGICAL PROPERTIES : DSR RESULTS

6% EVA -18 Blend

Neat bitumen



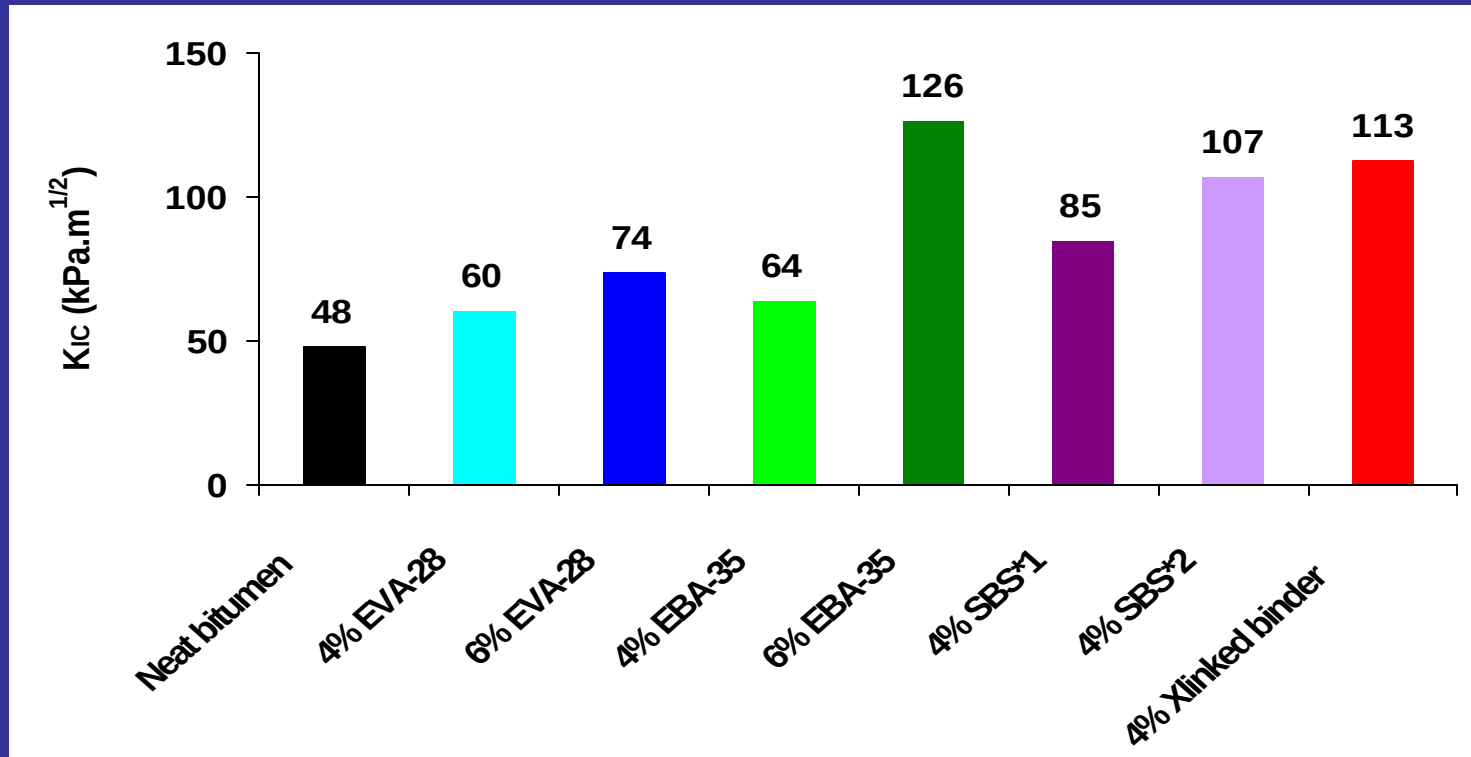
In both cases, polymer addition does not significantly improve the rheological properties of the binder at low temperature regardless of the polymer nature

RHEOLOGICAL PROPERTIES : DSR AND BBR

CONCLUSIONS

- Low temperature rheological properties of polymer modified bitumens seem to be controlled by the properties of the higher modulus phase :
⇒ *the bitumen matrix*
- *These conclusions could be different at higher polymer concentrations*
- *These results are not necessary valid for aged binders*

FRACTURE RESULTS AT -20°C



- Plastomer blends K_{IC} is slightly higher than the base bitumen
- 4% SBS* and SB mB's display higher K_{IC} than 4 % ethylene copolymer mB's
- 6% EBA needed to match 4% SBS* or SB mB's K_{IC}
- The 4% crosslinked binder displays a high K_{IC}
- Fracture properties significantly differentiate various polymer modification systems*

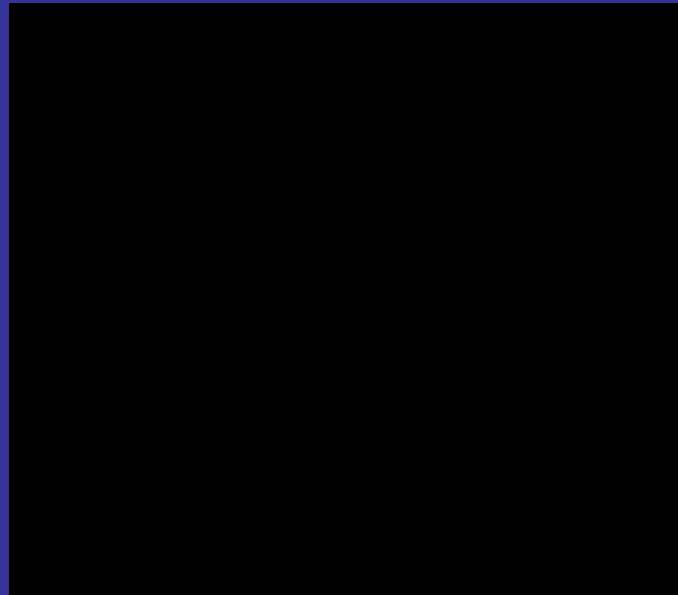
FRACTURE MECHANISMS

4 parameters have to be considered to explain the fracture mechanism

- **Binder morphology**
- **Volume fraction of the dispersed phase**
- **Adhesion between the phases**
- **Temperature difference between the testing temperature (-20°C) and the T_g 's of each binder component**

FRACTURE MECHANISM FOR A NEAT BITUMEN

$$K_{IC} = 48 \text{ kPa.m}^{1/2}$$



No topographic contrast

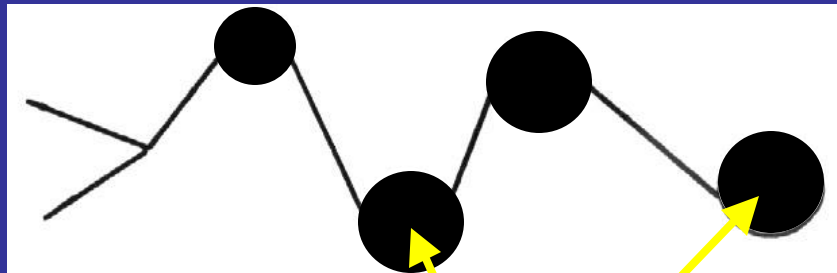
Brittle rupture $\Rightarrow$ low K_{IC}

MAIN FRACTURE MECHANISMS FOR PmB's

Crack deflection

The crack deviates from one particle to the next

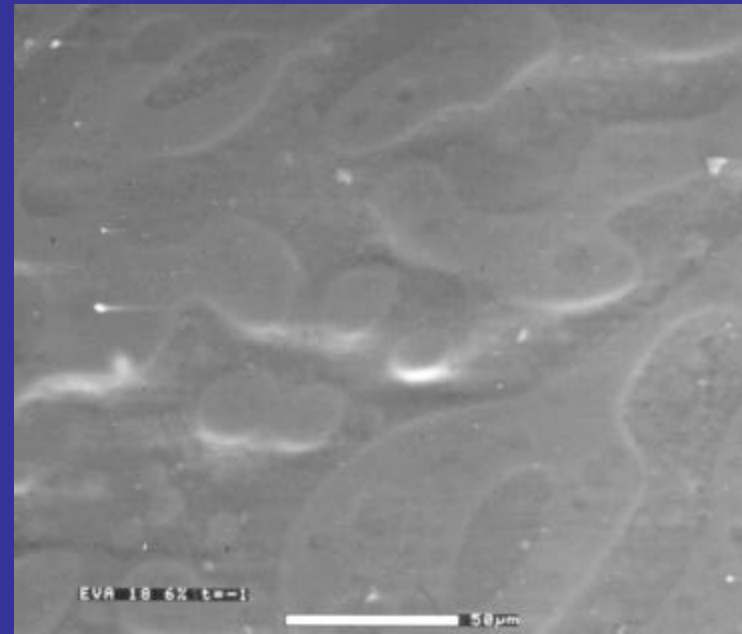
Particles are pulled-out



Polymer-rich nodules

CRACK DEFLECTION MECHANISM

6% EVA-28 blend
($K_{IC} = 74 \text{ kPa.m}^{1/2}$)



50 μm

EVA and EMA nodules are glassy at the testing temperature

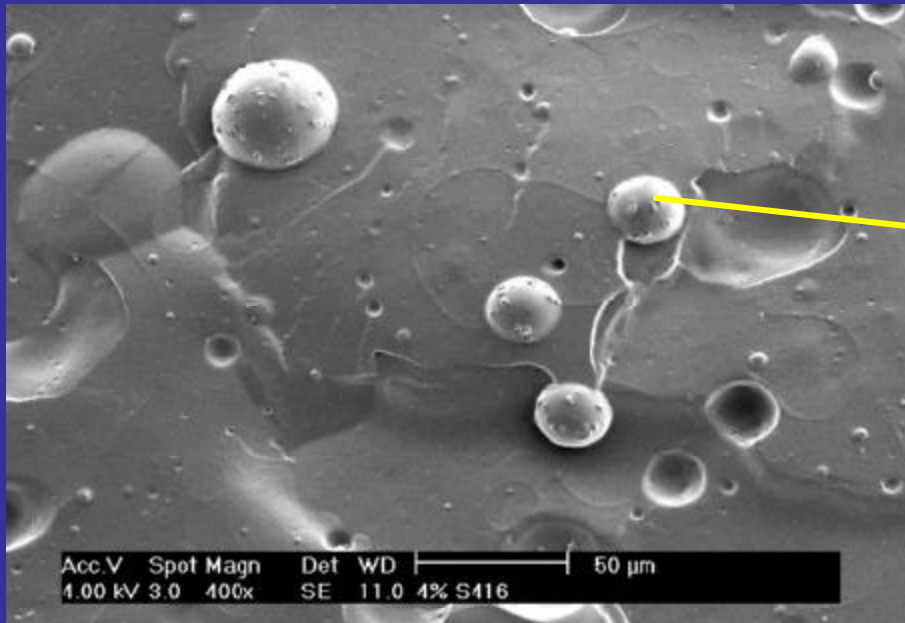
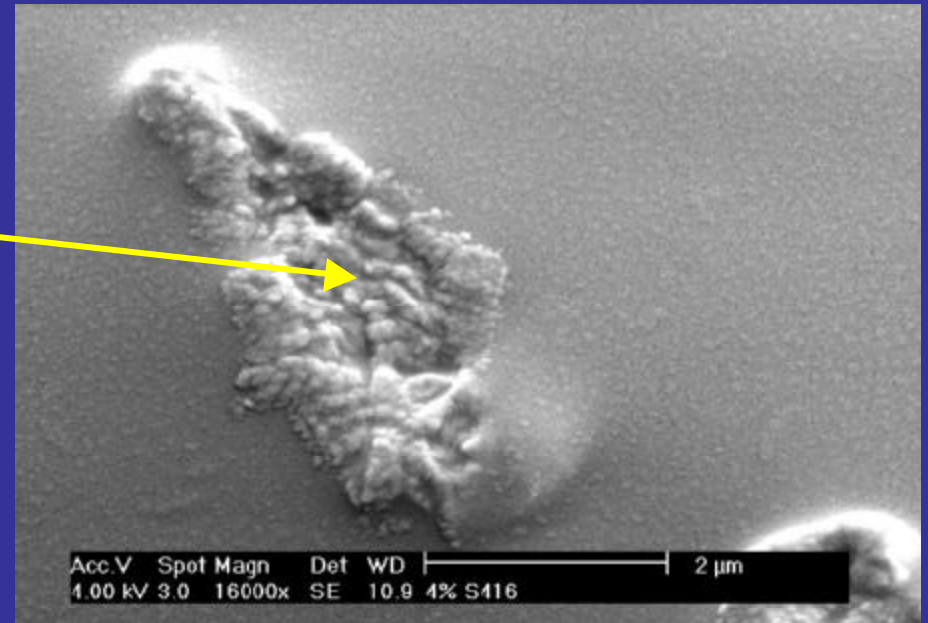
Polymer-rich particles are pulled-out without deformation

The fracture mechanism is governed by the (poor) adhesion between phases

⇒ **low K_{IC}**

CRACK DEFLECTION MECHANISM

*Blend based on 4% wt. SBS*₂ ($K_{IC} = 107 \text{ kPa.m}^{1/2}$)*



Fracture faces show particle pull-out (crack deflection)

Plastic deformation of polymer-rich nodules before being pulled-out

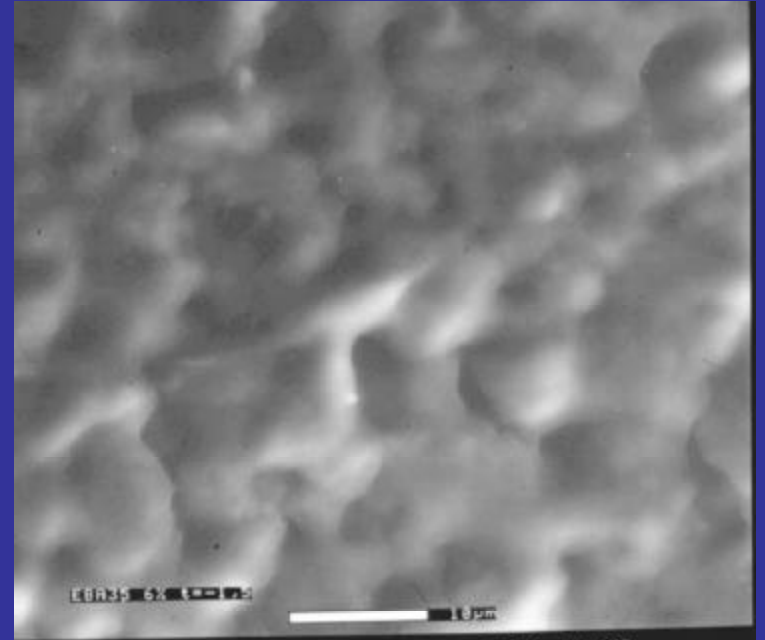
⇒ **High K_{IC}**

CRACK DEFLECTION MECHANISM WITH PLASTIC DEFORMATION

6% EBA-35 blend

$$K_{IC} = 126 \text{ kPa.m}^{1/2}$$

10 μm



EBA is in the rubbery state at -20°C

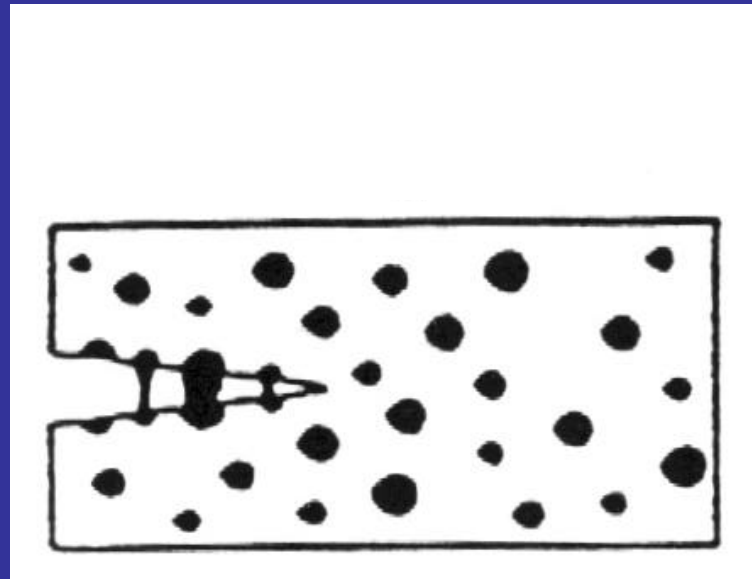
At 6%, the blend displays a finer co-continuous structure in favour of high toughness

$\Rightarrow$ **high K_{IC}**

CRACK FRONT BRIDGING MECHANISM

The crack propagates through the polymer-rich nodules which are stretched

The interfacial adhesion between the phases has to be high

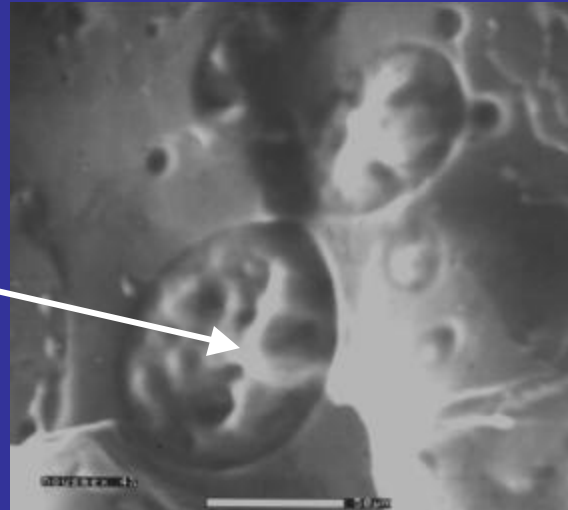


CRACK FRONT BRIDGING MECHANISM

4% SBS^*_1 blend

$K_{IC} = 85 \text{ kPa.m}^{1/2}$

**Stretching
of the
polymer-rich
nodules**



50 μm

The polymer-rich phase looks to be stretched (crack-bridging)

Good adhesion between phases

⇒ *high* K_{IC}

CRACK FRONT BRIDGING MECHANISM

4% Crosslinked binder

$$K_{IC} = 113 \text{ kPa.m}^{1/2}$$



Polymer-rich nodules are small

The dispersion of nodules is homogeneous

Fine interpenetration between the bitumen-rich matrix and polymer-rich domains

⇒ *High K_{IC}*

FRACTURE PROPERTIES : CONCLUSIONS

- **Polymer addition increases the bitumen fracture toughness at low temperature**
- **At low polymer content, the improvement is higher with elastomers than with plastomers due to :**
 - **different adhesion between phases**
 - **glass transition temperature**
 - **morphology of the blend**
- **These results need to be validated for mixes.**

SUMMARY

- Rheological properties at low temperature are mainly controlled by the bitumen matrix properties
- Contrary to experience, rheological tests do not discriminate between binders

SUMMARY (Cont'd)

- **Polymer addition increases the bitumen fracture toughness at low temperature**
- **At low polymer content, the improvement is higher with elastomers than with plastomers**
- **The fracture test allows to discriminate the low temperature thermal properties of asphalt binders**
- **Fracture mechanics is a good research tool, not a specification tool yet**

CONCLUSIONS (CONT'D)

- **More work is needed :**

- Effect of testing condition parameters need to be studied such as temperature with respect to T_g ,...
- These results need to be validated for asphalt mixes

- **Do not forget that**

- These conclusions are valid only for unaged binders
- These conclusions could be different for aged binders
- These conclusions could be different for highly modified binders

THANK YOU !

